

February 24-27, 2019

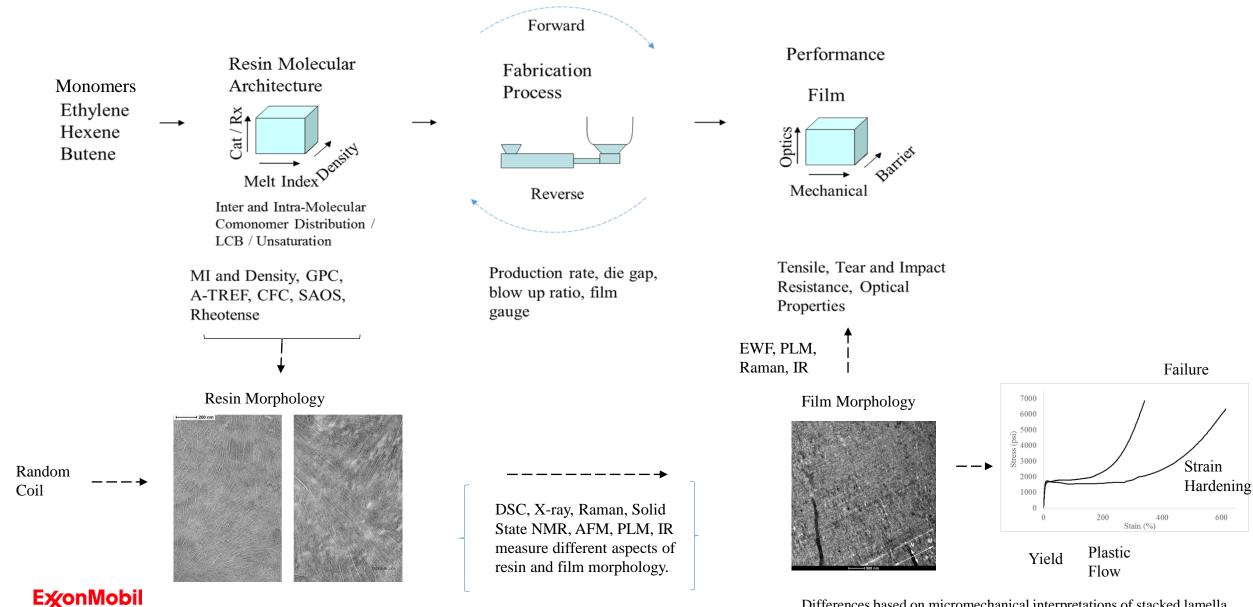
Developing Structure-Process-Property Relationships using Multivariate Analysis

Energy lives here

D. M. Fiscus

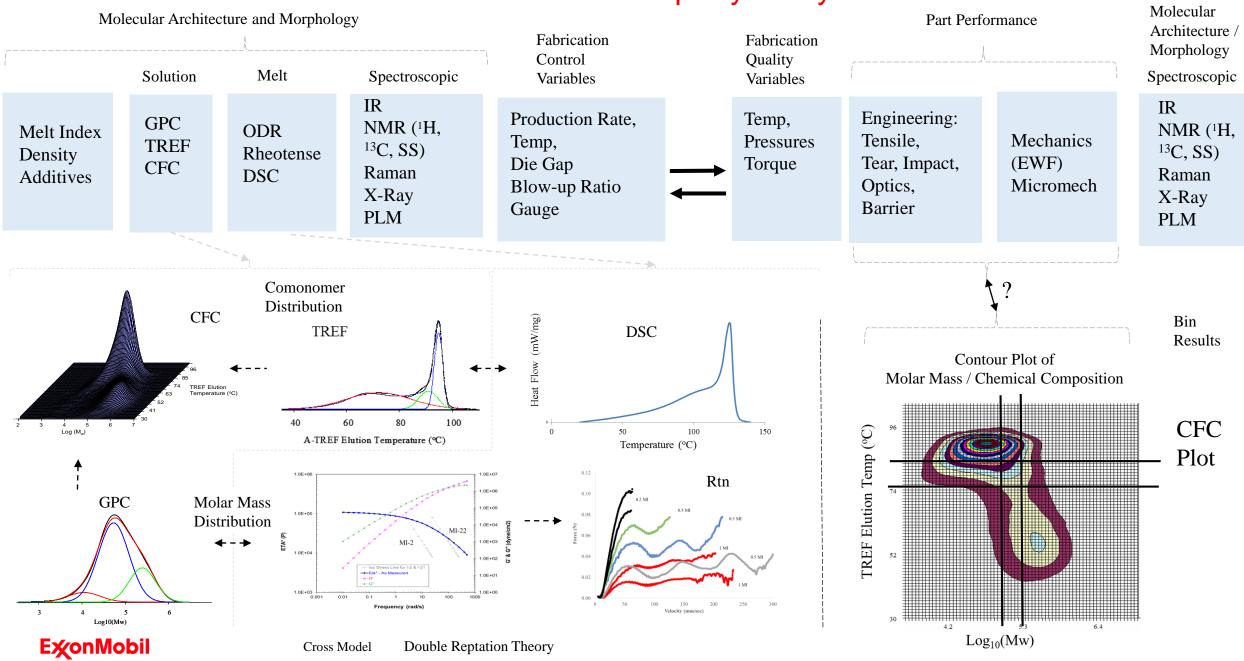
(David.M.Fiscus@ExxonMobil.com)

Overview



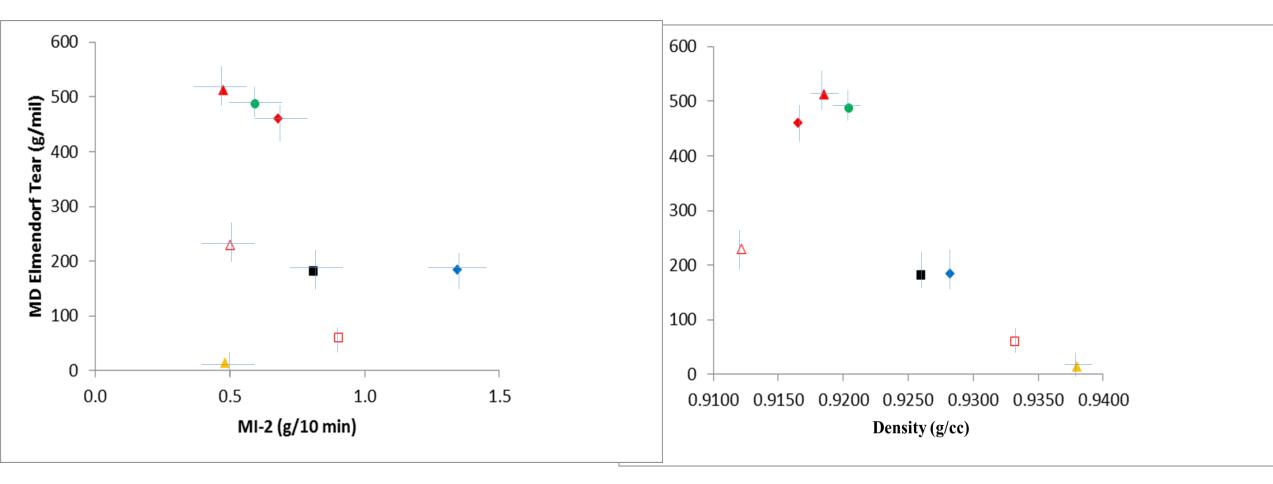
Differences based on micromechanical interpretations of stacked lamella. Extend to explaining toughness differences, like tear resistance.

Structure-Process-Property Analysis



Univariate Analysis

Relationships use different measurement scales. Relationships are general and not always monotonic.

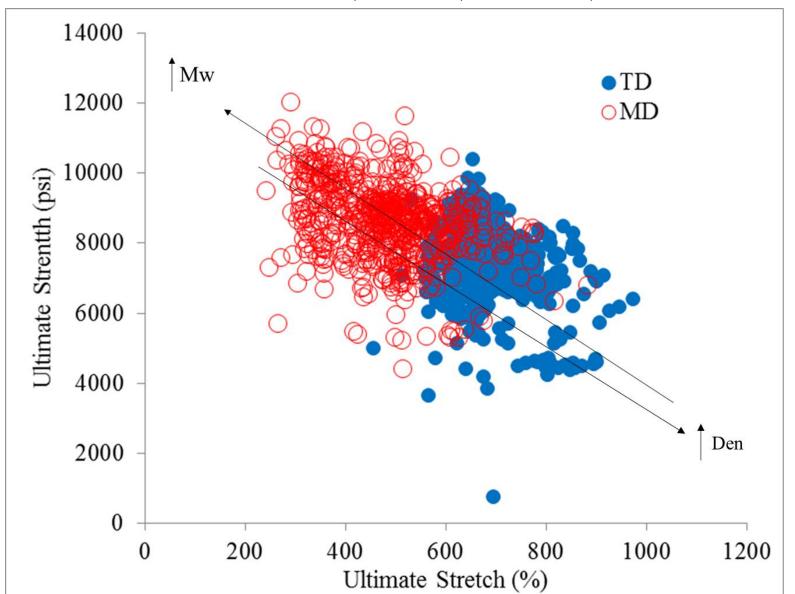


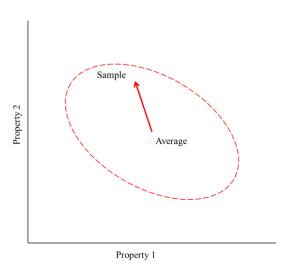
Average of selected samples.

E‰onMobil

Overlay Plot of Univariate Results

Liner PE Films; > 0.89 Den, 90-400 K Mw, Pd < 11



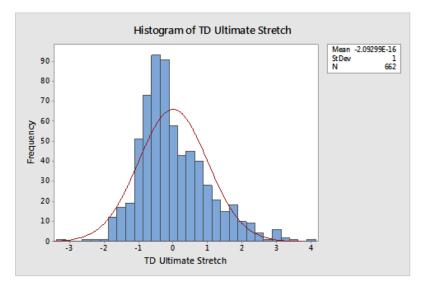


Statistical Analysis

Statistics removes differences due to measurement scales: variable effects are on equal footing.

Statistics describe populations

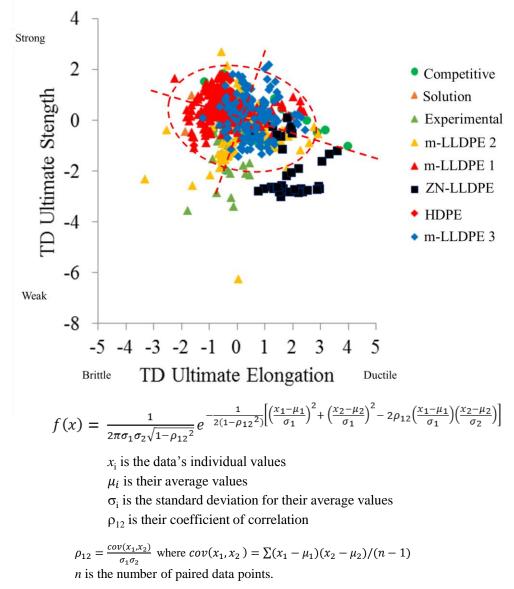
Performance increases with normalized values: Distance and direction depict performance



Mean \approx most frequent value Spread \approx four standard units of deviation Values are mean centered and scaled to unit variation.

$$f(x) = \frac{1}{\sqrt{2\pi\sigma^2}} e^{-\frac{(x-\mu)^2}{2\sigma^2}}$$

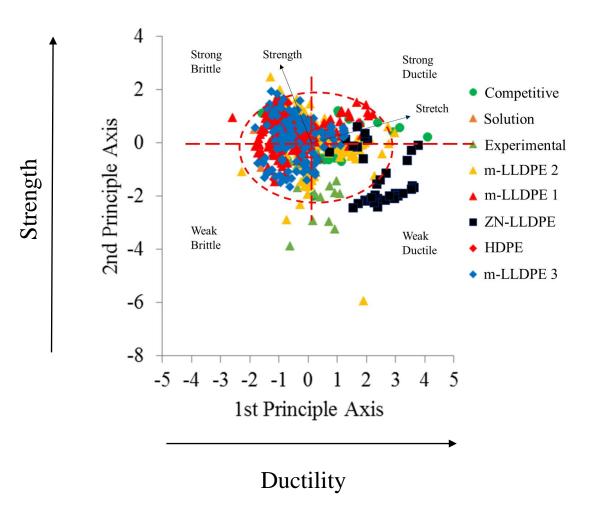
x is the individual data value μ is the data's average value σ is the standard deviation





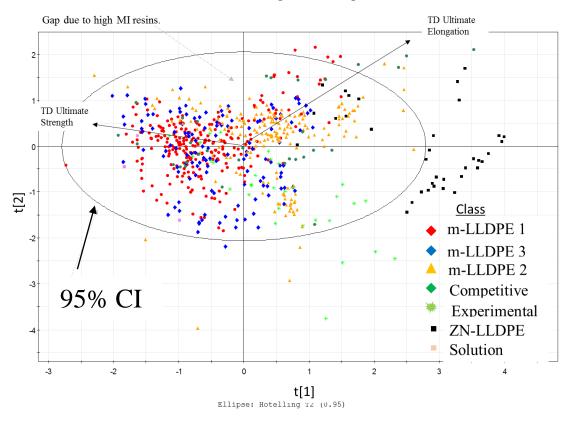
Dealing with Covariance

Principle axes reduces covariance.



Orthogonal axes eliminates covariance.

PCA for TD Ultimate Strength and Elongation



$$f(x) = \frac{1}{2\pi\sigma_1\sigma_2} e^{-\frac{1}{2} \left[\left(\frac{x_1 - \mu_1}{\sigma_1} \right)^2 + \left(\frac{x_2 - \mu_2}{\sigma_1} \right)^2 \right]}$$

 x_i is the individual data values

 μ_i is their average value

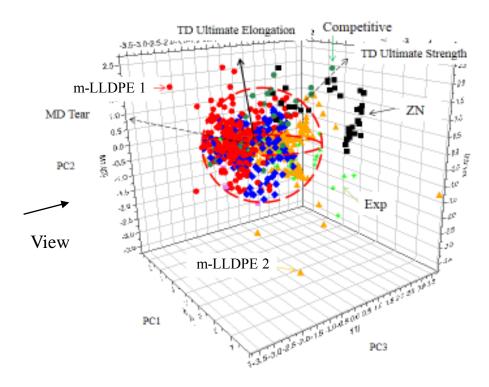
 σ_i is the standard deviation for those average values



PCA of MD Tear Resistance and TD Ultimate Properties

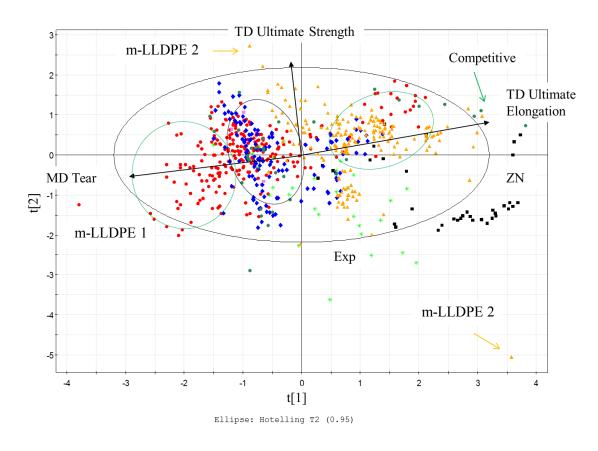
3D PCA plot

PCA Tear vs. Strength vs. Elongation T[Comp. 1]/t[Comp. 2]/t[Comp.3]



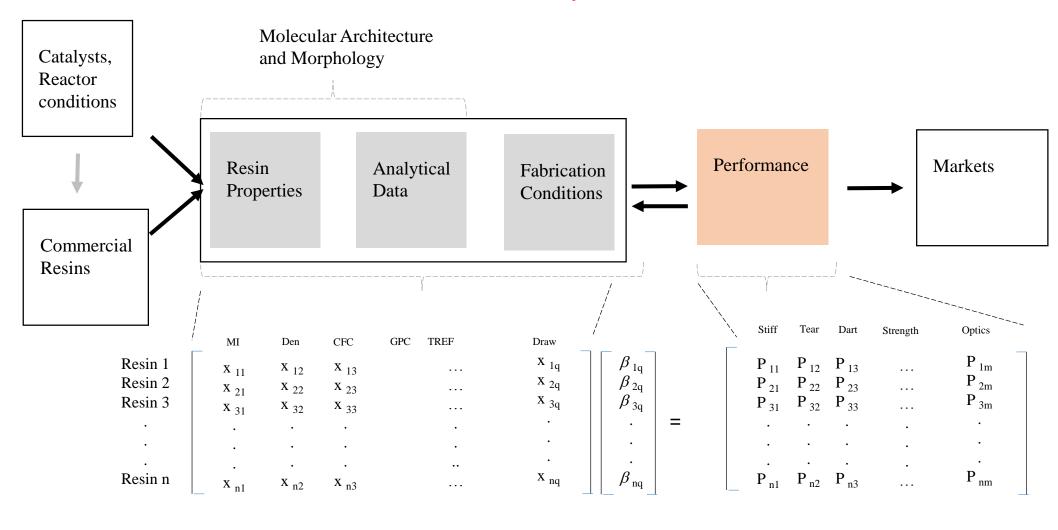
Projection onto coordinate planes gives bi-pots, like above.

PCA of MD Tear Resistance and Ultimate TD Performance





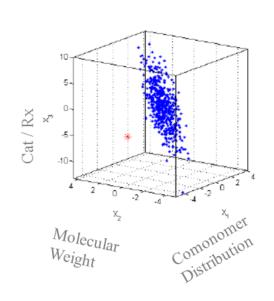
Process Map

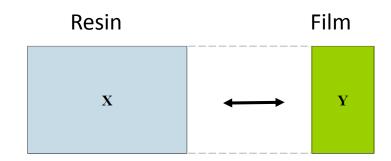


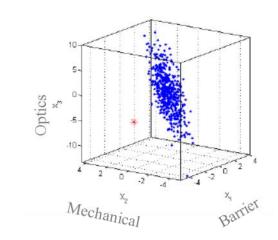


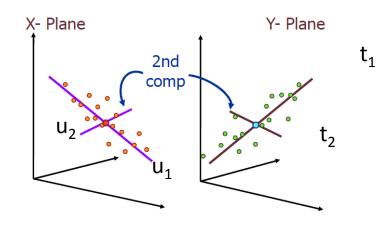
ExonMobil

PLS Background

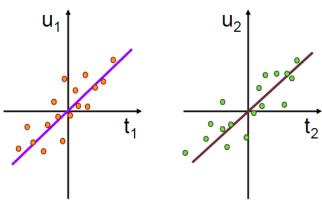




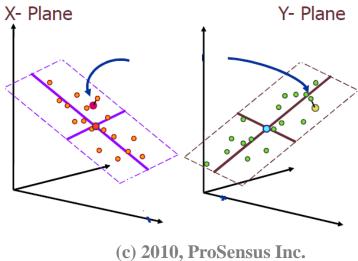




Principal Component Analysis

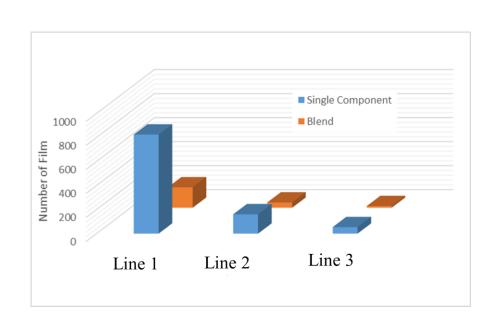


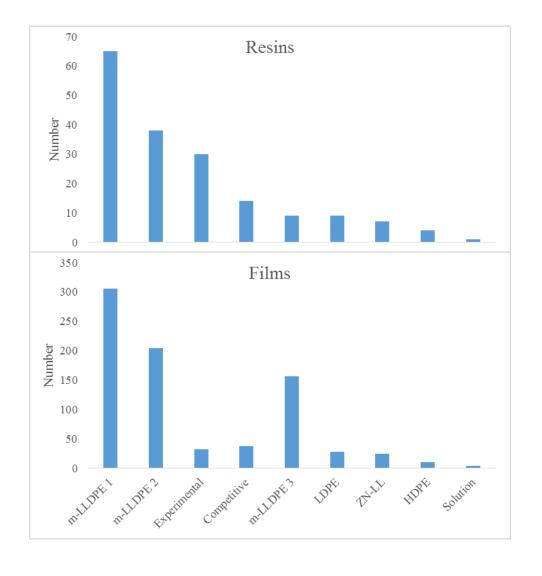
Principal Component Regression ExonMobil



+ 1 K Films by Resins and Line

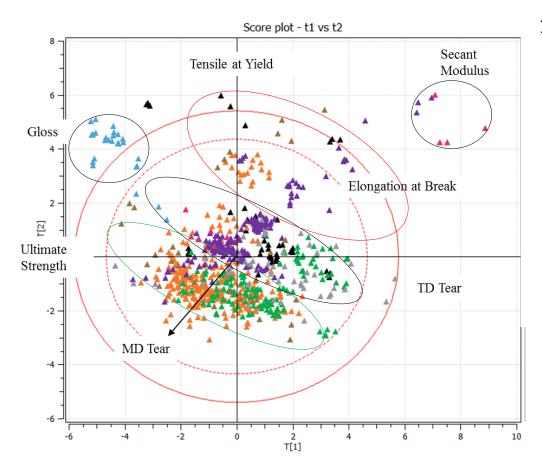
| Resin | Density | I2 | I21 |
|--------------|---------------|------------|------------|
| Family | (g/cc) | (g/10 min) | (g/10 min) |
| Competitive | 0.906 - 0.941 | 0.56 - 1.8 | 14 - 32 |
| m-LLDPE 1 | 0.904 - 0.933 | 0.35 - 1.3 | 8 - 34 |
| m-LLDPE 2 | 0.894 - 0.957 | 0.16 - 1.7 | 9 - 50 |
| m-LLDPE 3 | 0.903 - 0.928 | 0.98 - 1.9 | 15 - 32 |
| Experimental | 0.917 - 0.932 | 0.46 - 1.2 | 18 - 65 |
| ZN-LL | 0.918 - 0.944 | 0.81 - 3.1 | 20 - 88 |
| HDPE | 0.958 - 0.961 | 0.45 - 0.7 | 28 - 32 |
| Solution | 0.900 | 1.2 | |
| HP-LDPE | 0.926 - 0.951 | 0.35 - 2.5 | |





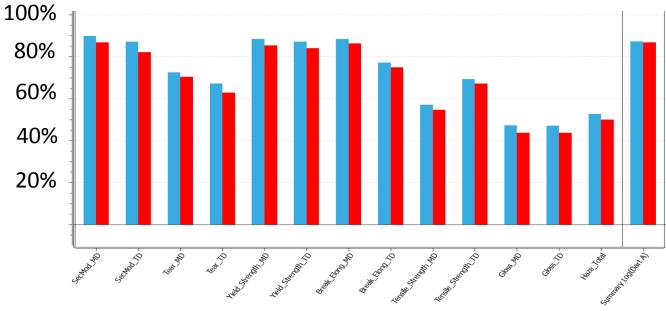
PCA for + 1 K Films

PCA Overlay Plot



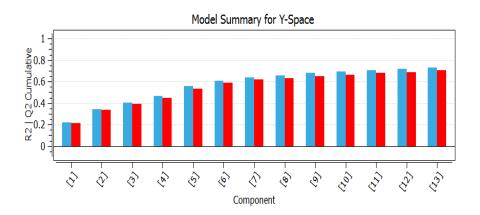
Explains (R^2 (cum)) and predicts (Q^2 (cum)) > 70% of the variation in the film's balance of properties.

 $(R^2(cum) \text{ and } Q^2(cum) \text{ by performance variable.}$



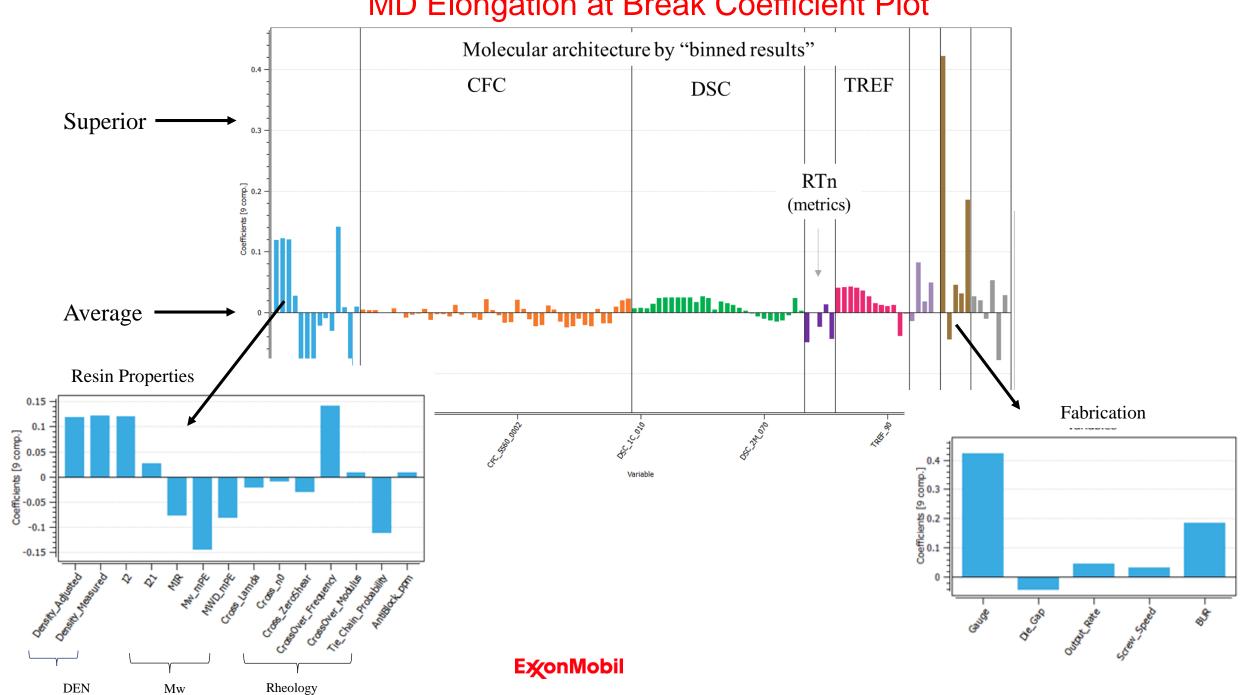
 R^2 (cum) are shown by blue bars.

Q²(cum) are shown by red bars, calculated through cross validation



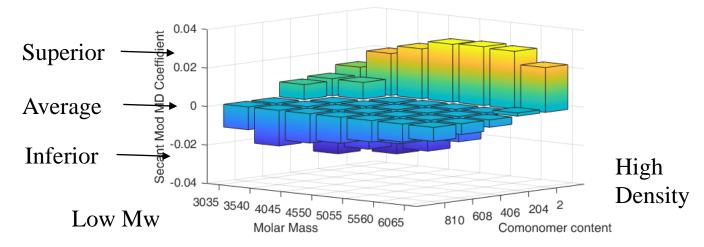


MD Elongation at Break Coefficient Plot

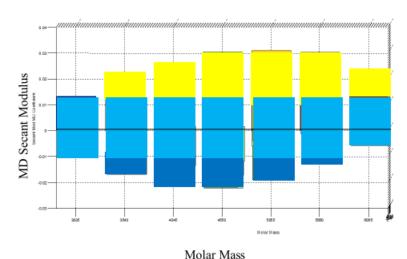


2D Coefficient Plots

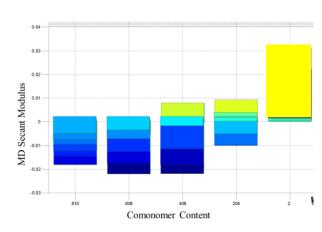
2D Coefficient Bar Graph MD Secant Modulus vs. Molar Mass and Comonomer Content



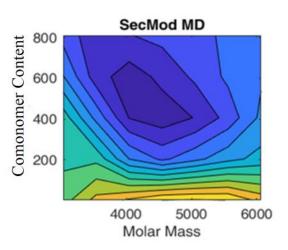
Coefficient Bar Graph MD Secant Modulus vs. Molar Mass



Coefficient Bar Graph
MD Secant Modulus vs. Comonomer Content



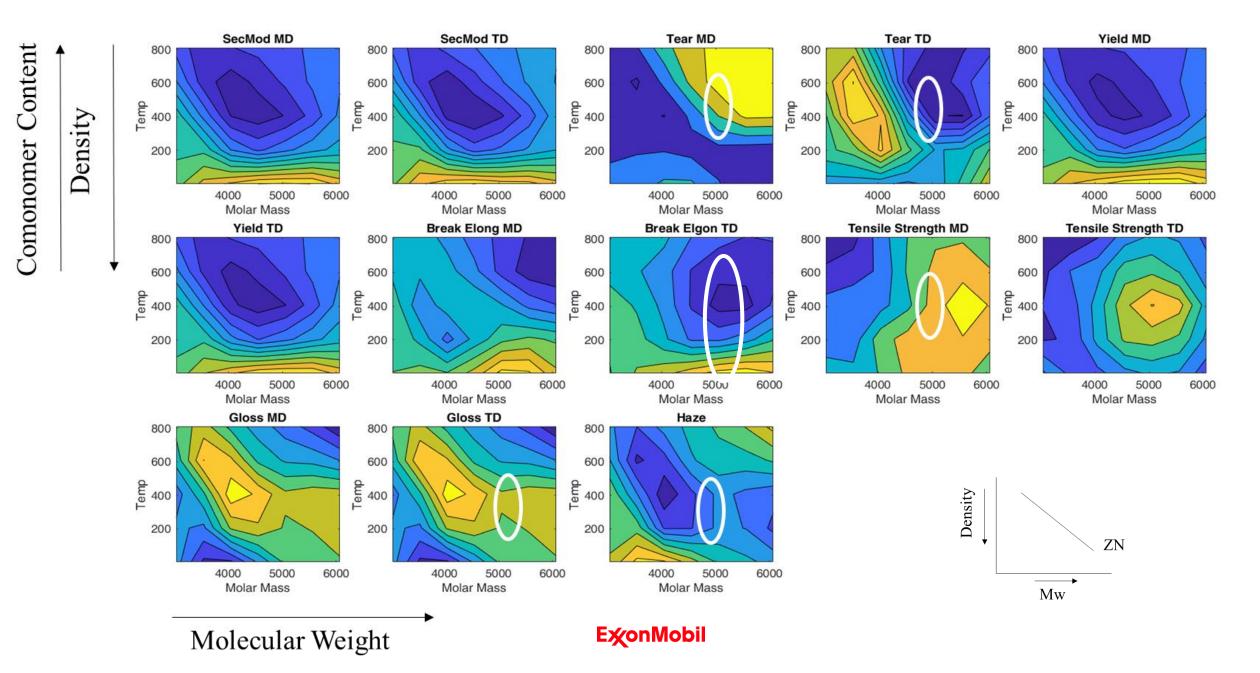
Contour Plot
MD Secant Modulus vs. Molar Mass and Comonomer Content



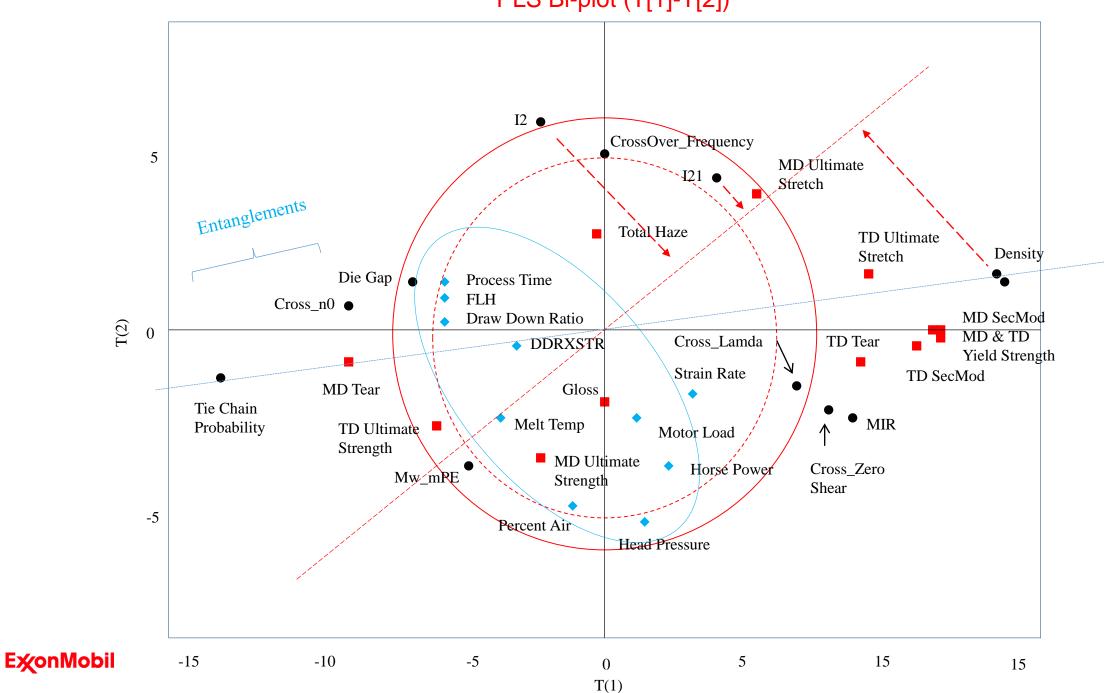


Coefficient increase with an increase in lighter color and decrease with an increase in darker color.

2D Coefficient Plots



PLS Bi-plot (T[1]-T[2])



Summary

Film performance depends on resin molecular architecture and film fabrication conditions.

Information on a wide variety of films using a broad spectrum of commercial resins was presented.

Structure-process-property relationships for blown film resins were established using multivariate statistical methods of analysis.

Optimum balance of film properties is obtained for molecular architecture.

Fabrication dependent and independent models show resin molecular architecture affects film performance more than film fabrication conditions.

- Fabrication independent models explain about 50% of the total variation in film performance.
- Fabrication dependent models explain about 70% of the total variation in film performance.
- 20% to 30% of film performance is unexplained by the fabrication dependent model.
- Application to resin blends was shown.

ExxonMobil is using this technology in developing new blown film resins with superior balances of performance.

Data generated by or on behalf of ExxonMobil

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Reduce Covariance Matrix into a Diagonal Matrix.

Linear Regression
$$y = a x + b$$

Mean Centered
$$(y - \overline{Y}) = a (y - \overline{Y})$$
 $a = \frac{Cov(x, y)}{Var(x)}$

$$a = \frac{Cov(x, y)}{Var(x)}$$

Extends to multivariate analysis

$$\begin{bmatrix} s_1^2 & s_{12} & s_{13} & \dots & s_{1m} \\ s_{21} & s_2^2 & s_{23} & \dots & s_{2m} \\ s_{31} & s_{32} & s_3^2 & \dots & s_{3m} \\ \vdots & \vdots & \vdots & \vdots & \vdots \\ s_{n1} & s_{n2} & s_{n3} & \dots & s_m^2 \end{bmatrix} \longrightarrow \begin{bmatrix} s_1^2 \\ s_2^2 \\ s_3^2 \\ \vdots \\ s_m^2 \end{bmatrix}$$

$$S \longrightarrow L$$

$$\begin{bmatrix} s_1^2 \\ s_2^2 \\ s_3^2 \\ \vdots \\ s_m^2 \end{bmatrix}$$

The covariance matrix is symmetric and non-singular. It is reduced using an orthonormal matrix U.

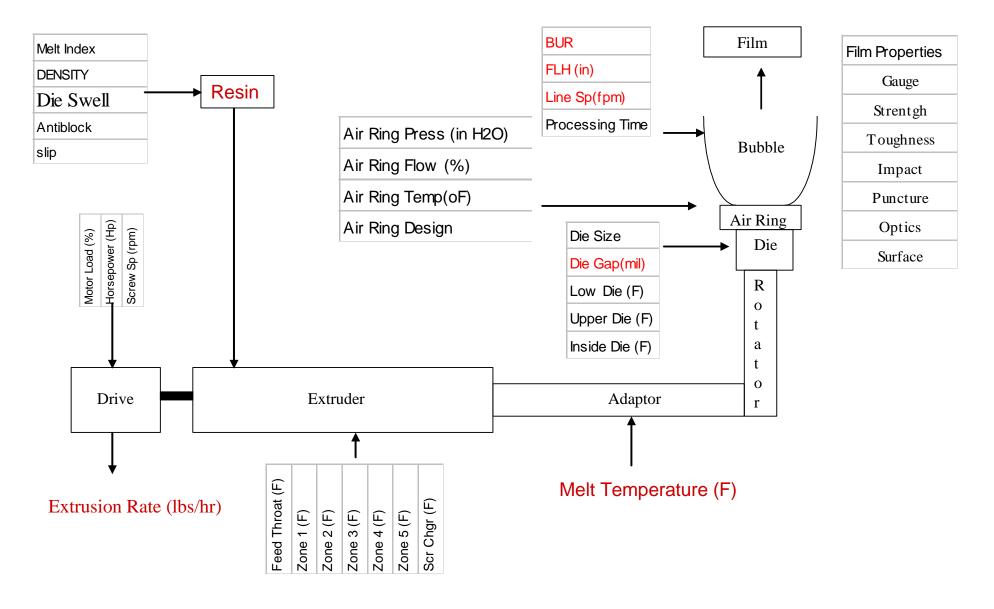
Where the components of U are the Eigen vectors of S, and those of L are the Eigen values of S. U'SU = L

 $|\mathbf{S} - l\mathbf{I}| = 0$ Solve U'SU for the Eigen values of S using the characteristic equation:

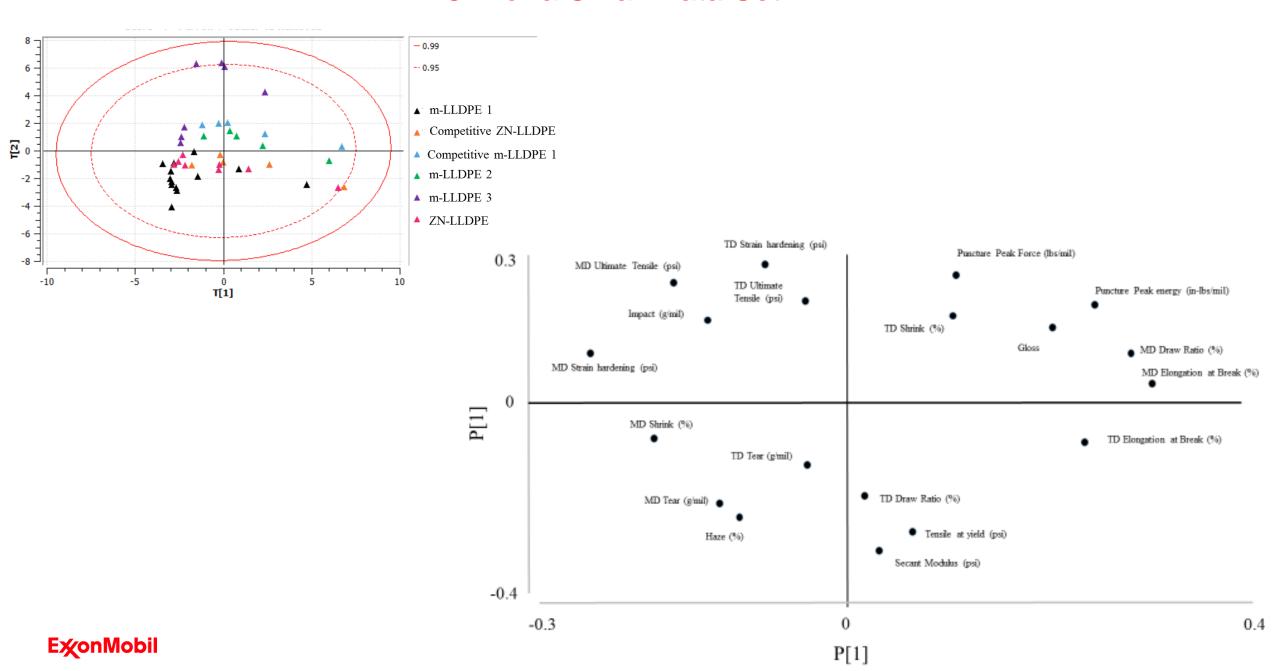
 $\begin{bmatrix} \mathbf{S} - l\mathbf{I} \end{bmatrix} \mathbf{t_i} = \mathbf{0}$ Use the Eigen values (u_i) to determine the Eigen vectors(l_i):

Technically, the Eigen vectors and values are determined iteratively using algorithms like NIPALS.

Blown Film Line Process Map



PCA for a Small Data Set



Test Methods

| Test Description | Method | Based on |
|----------------------------------|-----------------------|---------------|
| 1% Secant Modulus | ExxonMobil Procedure | ASTM D882-95a |
| Tensile Test | ExxonMobil Procedure | ASTM D882-95a |
| Elmendorf Tear | ASTM D1922-15 | |
| Dart Impact (Method A / B) | ExxonMobil Procedure | ASTM D1709-91 |
| Puncture Test** | ExxonMobil Procedure | ASTM D 5748 |
| Haze | ASTM D1003-13 | |
| Gloss | ASTM D618 | ASTM D2457-90 |
| Average Gauge Mic* | ExxonMobil Procedure | ASTM D374 |
| Resin Density | ASTM D1505-10 | |
| Melt Index | ASTM D1238-13 | |
| CFC Test | EM314-R4 | |
| SAOS (Shear Rheology) | EM513-R2 | |
| Rheotense (Extensional Rheology) | RHE04-3.3 & EM 514-R1 | |
| TEM (General TEM Analysis) | EM-116 R2 | |
| *Calibrated yearly | | |

^{*}Calibrated yearly.

DSC Analysis:

Samples of resin or film (3-5 mg) sealed in aluminum sample pans were heated and cooled in differential scanning calorimetry (DSC) at 10 °C/minute from -60 C to +220 C, from +220 C to -60 C and again from -60 C to +220 C with no hold time between each heating and cooling step. Each sample's heat of fusion (ΔH_f) was determined from its total heat flow (ΔH_f in J/g). Melting temperature was measured and reported during the second heating cycle (or second melt). Each sample's degrees of crystallinity was determined from its thermogram using the enthalpy of fusion of a perfect polyethylene crystal (ΔH_f^o) of 4110 J/mole. Common heats of fusion (ΔH_f^o) reported in the literature for a pure crystalline polyethylene ranges from 3977 J/mole to 4032 J/mole. The cumulative heat of fusion of each polyethylene was determined and the temperatures at 50%, 60%, 70% and 80% of the maximum cumulative heat of fusion noted.



^{**}A steal probe is used instead of a Teflon coated probe.

General References on Analytical Methods

Cross explains linear rheology

$$\eta_{\gamma} = \eta_{\infty} + \frac{\eta_0 - \eta_{\infty}}{1 + k \gamma^n}$$

 η_0 is zero shear viscosity k is a related to the relaxation time of the polymer in solution γ is shear rate n is the power law flow index

A. Yo. Malkin and A. I. Isayer, "Rheology: concepts, methods and Applications", ChemTec Publishing, Canada, 200

GPC explains linear rheology using Double Reptation Theory

$$G'(\omega) = 2G_{No} \times \sum_{i=0}^{N} w_i \times \sum_{j=i}^{N} w_j \times \frac{Z_i^2}{1 + Z_i^2}$$

$$G''(\omega) = 2G_{No} \times \sum_{i=0}^{N} w_i \times \sum_{j=i}^{N} w_j \times \frac{Z_i}{1 + Z_i^2}$$

$$Z_i = \omega \tau_e (M_i / M_e)^{3.4}$$

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